This article was downloaded by: On: 21 January 2011 Access details: Access Details: Free Access Publisher Taylor & Francis Informa Ltd Registered in England and Wales Registered Number: 1072954 Registered office: Mortimer House, 37- 41 Mortimer Street, London W1T 3JH, UK

International Journal of Polymer Analysis and Characterization

Publication details, including instructions for authors and subscription information: <http://www.informaworld.com/smpp/title~content=t713646643>

Computer-Aided Characterization of the Microstructure of Methyl Methacrylate-Ethyl Acrylate Copolymers by ¹³C-NMR Spectroscopy

F. Schmit-Quilèsª; M. Matlengiewiczʰ; D. Nicoleʰ a Université de Nancy I, Laboratoire d'Etude des Systèmes Organiques et Colloïdaux, Vandoeuvre-lès-Nancy Cedex, France **b** Polish Academy of Sciences, Institute of Coal Chemistry, Gliwice, Poland

To cite this Article Schmit-Quilès, F. , Matlengiewicz, M. and Nicole, D.(1996) 'Computer-Aided Characterization of the Microstructure of Methyl Methacrylate-Ethyl Acrylate Copolymers by ¹³C-NMR Spectroscopy', International Journal of Polymer Analysis and Characterization, 2: 2, 95 — 102

To link to this Article: DOI: 10.1080/10236669608233899 URL: <http://dx.doi.org/10.1080/10236669608233899>

PLEASE SCROLL DOWN FOR ARTICLE

Full terms and conditions of use:<http://www.informaworld.com/terms-and-conditions-of-access.pdf>

This article may be used for research, teaching and private study purposes. Any substantial or systematic reproduction, re-distribution, re-selling, loan or sub-licensing, systematic supply or distribution in any form to anyone is expressly forbidden.

The publisher does not give any warranty express or implied or make any representation that the contents will be complete or accurate or up to date. The accuracy of any instructions, formulae and drug doses should be independently verified with primary sources. The publisher shall not be liable for any loss, actions, claims, proceedings, demand or costs or damages whatsoever or howsoever caused arising directly or indirectly in connection with or arising out of the use of this material.

Computer-Aided Characterization of the Microstructure of Methyl Methacrylate-Ethyl Acrylate Copolymers by ¹³C-NMR Spectroscopy

F. SCHMIT-WILES*, M. MATLENGlEWlCZt **and** D. NICOLE*+

'UniversitB de Nancy I, Laboratoire #Etude des Systhmes Oganiques et Colloi'daux, URA CNRS 406, BP 239, F-54506 Vandoeuvre-/&-Nancy Cedex, France +Polish Academy of Sciences, Institute of Coal Chemist& Sowinskiego 5, PL-44100 Gliwice, Poland

(Received October 3, 1994; injnal form February 23, 1995)

¹³C-NMR spectra of methyl methacrylate-ethyl acrylate copolymers (PMMA/EA) offer sufficient resolution of **the carbonyl signal to observe pentad splitting. The triad distribution can be obtained by spectral simulation, while determination of pentad distribution requires additional semi-empirical calculation of their chemical shifts.**

KEY WORDS Methyl methacrylate, ethyl acrylate, copolymer, ¹³C-NMR, microstructure

INTRODUCTION

Previously, we had developed a method of computer-aided analysis of 13 C-NMR spectra of complex carbo- and petrochemical mixtures **[l].** The analysis was based on the comparison of chemical shifts and relative intensity data of the experimental spectrum with those collected in a library of pure compounds. In order to obtain information on individual lines in the overlapped zones of the spectrum, two deconvolution algorithms have been developed, based on the digital filtering and maximum likelihood principle **[2].** In the case of the analysis of polymer chain microstructure, the problem of superposition of a multitude of lines within a narrow spectral zone is frequently encountered. Despite the increase of the magnet field strength, which provides highly resolved spectra, the problem of overlapped lines still remains since the longer sequences are accessible. The longer sequences contain the precise description of the polymer chain. Moreover, the sequence distribution is usually governed by Bernoulli or Markov statistics, hence, even partial insight into the distribution of longer sequences provided by experimental data may permit the reconstitution of the parameters for the remaining sequences and subsequent ver-

⁺To whom all correspondence should be addressed.

Switzerland, May 24-26, 1994. Presented at the 7th International Symposium on Polymer Analysis and characterization, Les Diablerets,

ification of a particular model applied. The aim of this work is to show that it is possible to apply our method to the analysis of configurational-compositional sequence distribution of methyl methacrylate-ethyl acrylate copolymers.

EXPERIMENT

The industrial samples of the methyl methacrylate-ethyl acrylate copolymer (50/50), PMMA/EA, as well as the respective homopolymers: poly(methyl methacrylate), PMMA and poly(ethy1 acrylate), PEA, were synthesized at the Centre de Recherches Elf-Atochem (Carling, France), using azobisisobutyronitrile (AIBN) as a radical initiator; hence, the copolymer obtained was predominantly syndiotactic. The reaction was carried out in butan-2-one at 80°C for *5* h until complete consumption of the comonomers. The solvent was then evaporated under vacuum. The molecular weight of the copolymer was measured on a **GPC** apparatus in terahydrofuran solution and was found to be 50,100 g/mol.

The **100** MHz I3C-NMR spectra were recorded on a Bruker AM 400 spectrometer for 12 wt % (copolymer) and *5* wt % (homopolymers) solutions in CDC1, at 40°C. A reasonable signal-to-noise ratio was obtained after 1200 accumulations. In order to obtain quantitative spectra, the time delay between the pulses was five times greater than the longest longitudinal relaxation time [3]. Because $T_{\mu} = 3$ s, the delay was set to 15 s. The quantitative nature of the spectra was verified by comparison of the intensities for the carbonyl signal to the sum of those for α -CH₃ (in the methyl methacrylate units) and $OCH₂$ (in the ethyl acrylate units) which yielded good agreement (i.e., $I_{c=0} = 100.0$ and $I_{\alpha \text{CH3}(M)} + I_{O \text{CH3}(E)} = 100.2$). The composition of the copolymer was calculated from the ratio between the intensity of the $O-CH₃$ signal of ethyl acrylate and intensity of the carbonyl signals giving **48.4%** of ethyl acrylate, hence, 51.6% of methyl methacrylate. In the case of copolymers, the digital filtering algorithm was chosen for deconvolution of the carbonyl signals because of the presence of a great number of peaks [2].

METHODOLOGY

The analysis of the polymer microstructure was performed in four consecutive steps:

Recording of a quantitative ¹³C-NMR spectrum and assignment of the principal lines by comparison with the spectra of homopolymers, low-molecular-weight analogues, model compounds, etc..

Deconvolution of the overlapped lines. In the case of microstructure analysis, deconvolution seldom gives the parameters of all the individual lines. It should be regarded **as** a method yielding primary results for the most intense lines which can be later used as starting parameters for subsequent simulation and assignment.

Calculation of the line intensities. Deconvolution can provide the parameters, but for the major lines, it is generally possible to verify if these values follow a certain statistical model, usually Bernoulli or Markov of different orders. If so, the parameters of the remaining minor lines can be reconstituted. For the PMMAEA copolymer the values of the isotacticity parameters $\sigma_{MM} = 0.218$ for the methyl methacrylate units *(M)* and $\sigma_{EE} =$ 0.345 for the ethyl acrylate units (E) were calculated from the spectra of the respective homopolymers, while the co-isotacticity parameters, postulated to be indistinguishable [4], $\sigma_{EM} = \sigma_{ME} = \sigma^*$, were assumed to be equal to 0.41, according to San Román [5]. Since the copolymerization was performed until the complete disappearance of the comonomers, the first-order Markovian conditional probabilities for the compositional triads cannot be calculated from respective reactivity ratios since these calculations are valid only for low conversions; therefore, they were estimated from the NMR spectrum. Since

$$
P(E/M) = \frac{P(\overline{EM})}{2P(E)} \quad \text{and} \quad P(M/E) = \frac{P(\overline{EM})}{2P(M)},
$$

we had to determine $P(\overline{EM})$, where $\overline{EM} = EM + ME$ [6]. For acrylic polymers, only β -CH, signals can provide direct information on distibution of even sequences. We know from the respective homopolymers spectra that: (a) β -CH₂ diads from PMMA and PEA can be observed at **50.80-54.90** ppm and **3 1.15-36.90** ppm, respectively. The signals of ME diads are situated between these two regions. (b) The signals of α -CH (PEA), α -C and *0-CH,* (PMMA) are also present in the region between **40** and **52** ppm and their lines overlap with those of *PCH,* signals (Figure 1). Therefore, in order to obtain the intensities of respective β -CH₂ signals by simple integration, it is necessary to substract the intensities of intervening signals calculated from the signals outside this region (since $I_{OCH2} = I_{\alpha\text{-}CH}$ for PEA and $I_{\alpha\text{-}CH3} = I_{\alpha\text{-}C} = I_{OCH3}$ for PMMA). P(E/M) and P(M/E) were found to be equal to **0.62** and 0.58, respectively. we had t
CH₂ sign
from the
can be o
ME diad

Simulation of a copolymer spectrum. To simulate a copolymer spectrum, the intensities and the line positions are used as starting parameters of the simulation algorithm using the Lorentzian line-shape model. As stated above, the intensity values can be derived from deconvolution and subsequent verification of a particular statistical model, while the positions of the lines have to be estimated separately **[7].** They are usually obtained by calculations of the possible different statistical configurations (γ gauche effect [8,9]) or semiempirical incremental calculations [**10,111.**

FIGURE I copolymer. 100 **MHz** ¹³C-NMR spectrum of the region containing the β -CH₂ signals of the PMMA/EA

RESULTS AND DISCUSSION

The methodology presented here for a copolymer has been already positively tested for a sample of a model homopolymer, syndiotactic PMMA [12]. The carbonyl signal if this sample in the 100 MHz ¹³C-NMR spectrum exhibited traces of the heptad lines. Application of the deconvolution with the maximum likelihood algorithm can provide deeper insight into the heptad distribution using first-order Markov statistics.

The carbonyl region of the PMMA/EA copolymer and respective PMMA and PEA homopolymers is presented in Figure 2. It can be seen that strong configurational effect is introduced by the M units, while E units give rather slight effects. At first, assignment of the lines at the triad level can be accomplished based on the results published by Cheng [13] and San Román and Valero [5] who have assigned 12 triads considering only the conformation effect of the M unit.

FIGURE 2 100 MHz ¹³C-NMR spectra of the carbonyl signal of the PMMA/EA copolymer (a), PMMA (b) and PEA (c) homopolymers. Lines assignments according to San Román [5].

MICROSTRUCTURE OF METHYL METHACRYLATE ETHYL ACRYLATE COPOLYMERS 99

From these data, the simulation of the carbonyl spectrum of the copolymer with 12 **Hz** linewidths can provide substantial agreement between the experimental and simulated spectra. The results of these simulations (from data of Table I) performed according to different statistical models for configurational and compositional sequences, that is, for the 20 theoretically triads, show that the Markov-Bernoulli model (Figure 3b) fits better than the Bernoulli-Bernoulli model (Figure 3a). Since the investigated industrial copolymer sample was prepared by high-conversion polymerization, the values of the isotacticity parameters taken from homopolymer measurements and especially the compositional conditional probabilities calculated from the 13C-NMR spectrum were not quite adequate; there was visible discrepancy for the outermost signals of the two homotriads. Moreover, according to the assignments proposed by San Román and Valero [5] there was significant discrepancy between the lines at about 174.9 and 175.9 ppm, ascribed to be the ErMrE and $[EME + MrMmE + MmMmE]$ triads, respectively. If we reverse these two lines instead, we arrive at a much better fit which suggests the change in the assignment. The new assignments are presented in Figure 3c.

Nevertheless, it is obvious that precise pentad assignment is essential to obtain better agreement between the experimental and the simulated spectra, since numerous pentads having identical central triad can significantly overlap and contribute to different triads. It indicates that a more sophisticated statistical model could be necessary **to** further increase the precision of the sequence distribution determination. Theoretically, there are 272 pentads observable in the **NMR** spectrum for a copolymer which reflect both configurational and sequence effects. The spectrum resolution of the carbonyl signal recorded at 100 MHz makes it possible to observe the pentad splitting; nevertheless, the overlapping of pentad lines is so prevalent that deconvolution can yield not more than 100 lines (Figure **4).**

ASSIGNMent of the resonance pattern or carbonyl carbons.				
Chemical shift	Copolymer sequence	Bernoulli/ Bernoulli	Bernoulli/ Markov first order	Experimental (deconvolution)
177.87	MrMrM	0,0703	0.0453	0.0920
177.20	MrMmM	0.0393	0,0253	0.0266
176.87	MrMrE	0.1120	0,1068	
176.81	MrMmE	0,0779	0,0742	
176.68	MmEmM	0,0204	0,0288	0.2998
176.53	MmErM	0.0588	0,0829	
176.38	MrErM	0.0423	0,0596	
176.22	MmMmM	0.0055	0,0035	0.0151
176.05	EmMmE	0,0216	0,0304	
175.90	EmMrE	0.0621	0,0875	0.1618
175.80	ErMrE	0,0447	0,0630	
175.65	MmEmE	0,0360	0,0352	
175.55	MrEmE	0.0518	0.0506	0.1699
175.47	MmErE	0,0692	0,0676	
175.35	MrErE	0,0996	0,0972	
174.90	MmMrE	0,0313	0,0298	
174.80	MmMmE	0.0218	0,0207	0.0972
174.48	EmEmE	0,0159	0.0107	
174.41	ErEmE	0,0610	0,0413	0.1376
174.36	ErErE	0,0586	0,0396	

TABLE 1

Calculated and experimental (see Figure 4) probabilities for methyl methacrylate ethyl acrylate copolymer. Assignment of the resonance pattern of carbonyl carbons.

FIGURE 3 Simulation of the PMMAEA carbonyl signal at the triad level.

a) Bernoulli distribution for both configurational and compositional triads

b) First-order Markov model for compositional triads and Bernoulli model for configurational triads and assignments according to San Román [5] and Cheng [11].

c) First order Markov model for compositional triads and Bernoulli model for configurational triads, but with reversed assignments for the EME and MmME sequences.

FIGURE **4 Deconvolution of the PMMA/EA carbonyl signal by means of the digital filtering algorithm.**

Moreover, the deconvolution results yield many lines larger than a single line, indicating that several groups of two, and three peaks are not resolved into individual lines. Therefore, deconvolution can provide only the starting parameters for separate calculations of line intensities (verification of different statistical models) and positions (incremental calculations of chemical shifts **[10,1** l]), which is currently being investigated.

CONCLUSIONS

The method developed for the analysis of complex petrochemical mixtures has proved useful for the analysis of copolymer microstructure. Also, the spectrum simulation at the triad level can provide fast results for **an** approximate analysis. Complete distribution of the configurational-compositional pentads can be achieved by spectral simulation which necessities accurate data for 272 lines. FuIl pentad analysis, in progress, can result in establishing a library of their chemical shifts that can be very useful in rapid determination of sequence distribution at **pentad** level for samples of **PMMA/EA** copolymers of different tacticity **and** composition.

Acknowledgments

The authors are grateful to the "Programme International de Coopération Scientifique CNRS - Académie des Sciences de Pologne" for financial support. All NMR **spectra** were **recorded on** the spectrometers of the NMR Center of the University of Nancy I.

References

- **1.** M. Matlengiewicz, N. Henzel. J.-C. Lauer, T. Laurens, D. Nicole, and P. Rubini, *Anafyst.* **117, 387 (1992).**
- 2. **M.** Matlengiewicz, N. Henzel, D. Czachowska, **F.** Schmit-Quiks, D. Nicole, and J.-C. Lauer, Fuel, **73, 843 (I 994).**
- **3.** K. Hatada, T..Kitayama. Y. Okamoto, **K.** Ohta, **Y.** Umemura, and H. Yuki, *MakromoZ, Chem* **179, 485 (1978).**
- **4.** H. Yuki, Y. Okamoto. Y. Shimada, **K.** Ohta, **and** K. Hatada, *J.* Polym *Sci.. Polym Chem Ed.* **17, 1215 (1979).**
- 5. **J. San Román and M. Valero,** *Polymer***, 31**, 1216 (1990).
- **6.** E. Klesper and G. Sieglaff, In *Polymer Spectroscopy;* D. *0.* Hummel, Ed., Weinheim, Verlage Chemie, **1974.**
- **7.** *H.* N. Cheng, *Trendr Anal. Chm* ., **13.95 (1994).**
- **8. A.** E. Tonelli, *Macromolecules, 24,* **3065 (1991).**
- **9.** T. **Asakura,** M. Demura, **and** Y. Nishiyama, *Macromolecules,* **24,2334 (1991).**
- **10. A.** Johnsen, E. Klesper, and T. Wirthlin, *Macromol. Chem.,* **177.2397 (1976).**
- **11.** M. Matlengiewicz, *Macromolecules,* **17,473 (1984).**
- **12.** M. Matlengiewicz, submitted for publication in *Polymer.*
- 13. H. N. Cheng, *J. Chem In\$ Compur. Sci.,* **27,8 (1987).**